# **Diamond formation and behaviour of carbides in several 3d-transition metal-graphite systems**

SHIGEHARU NAKA, AKIHIRO TSUZUKI, SHIN-ICHI HIRANO *Synthetic Crystal Research Laboratory, Faculty of Engineering, Nagaya University, Fiuo-cho, Chikusa-Ku, Nagaya, Japan* 

The formation of diamond and the behaviour of coexisting carbides with diamond in several 3d-transition metals (manganese, iron, cobalt, nickel)-graphite systems was studied under 7 GPa pressure up to  $1700^\circ$  C from the view-point of diamond formation. In the nickel-graphite and cobalt-graphite systems, no stoichiometric carbide was formed, but  $M<sub>x</sub>C$ , which is thought to be the interstitial solid solution of carbon, was formed. In the iron-graphite system, on the other hand, the formation of two stoichiometric carbides, i.e. Fe<sub>3</sub>C and Fe<sub>7</sub>C<sub>3</sub>, was found. In the manganese-graphite system, only one stoichiometric carbide  $Mn<sub>7</sub>C<sub>3</sub>$ , which is isostructural with Fe<sub>7</sub>C<sub>3</sub>, was formed. The diamond formation process in the presence of these transition metals is discussed in relation to the carbide formation.

## **1. Introduction**

Diamond has been synthesized from graphite using some transition metals as a solvent-catalyst. In general, solvent-catalyst metals have been classified into two categories. The first is the metals which form no stoichiometric carbides under the condition for diamond synthesis. The second is the metals which do form stoichiometric carbides. The former consist of nickel, cobalt, platinum, etc. and the latter of iron, manganese, chromium, etc. Giardini and Tydings [1] proposed a relation between diamond formation and carbides. They supposed that diamond was simply precipitated from supersaturated solution with carbon in the nickel-graphite system and was associated with the formation of several stoichiometric carbides in the cobalt-graphite system, They, on the other hand, proposed in the iron or manganese-graphite system that the stoichiometric carbide with higher carbon content decomposed into the carbide with lower carbon content to precipitate diamond.

There have been strong doubts about the diamond formation mechanism in relation to the stability of the coexisting carbide, especially in the system where stable stoichiometric carbides are formed. In the present work, the formation of diamond and coexisting carbides are investigated systematically by use of a 3d-transition metal (nickel, cobalt, iron or manganese) in order to elucidate whether the diamond can be formed by the solution-precipitation process or the decomposition of the carbide.

### **2. Experimental details**

The high temperature-high pressure experiments were carried out with a girdle-type high pressure apparatus. Fig. 1 shows the arrangement of the pressure cell. The pressure was calibrated at room temperature by the transition points of  $Bi_{I-II}$  $(2.55 \text{ GPa})$ ,  $T_{\text{II}-\text{III}}$   $(3.67 \text{ GPa})$ ,  $Ba_{\text{I}-\text{II}}$   $(5.5 \text{ GPa})$ and  $Bi_{III-V}$  (7.7 GPa). The temperature was calibrated under 7 GPa pressure by the melting points of silver and gold, which were fixed according to the pressure effect on the melting point [2]. The estimated errors of pressure and temperature were  $\pm$  0.3 GPa and  $\pm$  50 $\degree$  C, respectively.

The starting materials were well crystallized graphite powders prepared from pitch cokes at  $2800^{\circ}$ C and highly pure metal powders of nickel,





*Figure l* High pressure cell arrangement.

cobalt, iron and manganese. The mixture of 80 wt % graphite and 20 wt % metal (about 95 at % carbon and 5at% metal) was placed in the pressure cell. The pressure was raised to 7 GPa and then the whole of the girdle-type pressure cylinder was kept cool in a water jacket. The temperature was increased to the desired value by passing the electric current through a glassy carbon heater. After holding for 10min, the electric current was cut off and the sample was quenched down to room temperature. The treated specimens were identified by an X-ray diffraction method and the magnetic properties of the specimen were also measured. The relative amounts of the crystalline phases were determined by the calibration curve of the X-ray diffraction intensities, which was obtained beforehand (by the correction for the preferred orientation of the crystalline phases to the relationship between the X-ray diffraction intensities) with a certain amount of crystalline phase in the mixture. The X-ray diffraction lines used for the measurement of the relative intensity were 1 1 1 for diamond and 0 0 2 for graphite.

#### **3. Results and discussion**

#### 3.1. Diamond formation in nickel-graphite system

In the nickel-graphite system, no stoichiometric carbide was observed in agreement with the previous report [1 ] and nickel interstitial solid solution of carbon (fcc type) was found to form. It was reported that the diamond syntheses by use of nickel include the carbide Ni<sub>x</sub>C ( $x > 4$ ) which has a larger lattice constant than that of nickel metal itself [3]. Fig. 2 shows the changes in the relative amounts of crystalline phases in the treated speci-

*Figure 2* Changes in relative amounts of crystalline phases formed by the mixture of 80 wt% graphite-20 wt% nickel under 7 GPa pressure for 10 min: diamond  $(e)$ ;  $\text{Ni}_{x}C$  (o).

men with increase in the heat-treatment temperature at 7 GPa pressure. The amount of  $Ni_xC$  was not changed up to  $1700^{\circ}$  C. The diamond could be detected at  $1400^{\circ}$ C and its amount increased rapidly with increase in temperature. The yield of diamond was above 70 wt % of the sample above 1500 $^{\circ}$ C. The change of lattice constant of Ni<sub>x</sub>C is shown in Fig. 3. The lattice constant increased gradually up to  $1400^{\circ}$ C and steeply increased at higher temperature, followed by the approach to the constant value at  $1700^{\circ}$  C.

The nickel reacts with graphite to form low carbon content carbide  $Ni_xC$  (with large x) at lower temperature and, above the nickel solid solution-diamond eutectic temperature, diamond



*Figure 3* Change of lattice constant  $(a_0)$  of Ni<sub>x</sub>C with heat-treatment temperature.





*Figure 4* Changes in relative amounts of crystalline phases formed by the mixture of  $80$  wt% graphite-20 wt% cobalt under 7GPa pressure for 10min: diamond **(e);**   $\alpha$ -cobalt ( $\circ$ ); Co<sub>x</sub>C ( $\circ$ ).

is formed from supersaturated solution, as suggested by Strong and Hanneman [4] from the nickel-carbon phase diagram under 5.4GPa pressure. On quenching, the same carbon content  $Ni<sub>x</sub>C$  is solidified. The nonstoichiometric carbide  $\text{Ni}_{x}C$  is ferromagnetic and has lower Curie temperature than nickel metal itself.

#### 3.2. Diamond formation in cobalt-graphite system

As in the nickel-graphite system, no stoichiometric carbide was detected in the cobalt-graphite system and  $\beta$ -cobalt interstitial solid solution of  $Co<sub>x</sub>C$  (fcc type) was formed. Fig. 4 shows the relative amounts of the crystalline phases with increase in the heat-treatment temperature. At low temperature, a trace amount of  $\alpha$ -cobalt (hcp) coexisted with the fcc type of  $Co_xC$ , but it disappeared when a large amount of diamond was formed. The diamond formation began at  $1400^{\circ}$ C and its amount increased abruptly with increase in temperature. The yield of diamond was above 70 wt % in the specimens treated above  $1500^{\circ}$  C. The change of lattice constant of  $Co<sub>x</sub>C$  is shown in Fig. 5. The change was similar to that in the case of  $Ni<sub>r</sub>C$ .

These results indicate that  $Co_xC$  is formed by reaction of cobalt with graphite at low temperature, and that diamond is formed from the supersaturated solution, as in the nickel-graphite system.  $\alpha$ -cobalt was found to disappear when the epitaxial growth of  $Co_xC$  took place on diamond during the solidification. The lattice constant of

*Figure 5* Change of lattice constant  $(a_0)$  of Co<sub>x</sub>C with heat-treatment temperature.

 $Co<sub>x</sub>C$  above 1500°C was close to that of diamond. This fact might reflect the superior wettability of cobalt in the liquid phase sintering of diamond powder [5]. It was not possible to measure the Curie temperature of  $Co_xC$  in this work up to  $900^{\circ}$ C, but its value was expected to be lower than that of the cobalt metal itself.

#### 3.3. Diamond formation in iron-graphite system

In the iron-graphite system, two stoichiometric carbides, i.e.  $Fe<sub>3</sub>C$  and  $Fe<sub>7</sub>C<sub>3</sub>$ , were formed. Fig. 6 shows the relative amounts of the crystalline phases with increasing temperature. Iron reacted with graphite to form  $Fe<sub>3</sub>C$  at around 1000°C,



*Figure 6* Changes in relative amounts of crystalline phases formed by the mixture of 80 wt % graphite-20 wt % iron under 7 GPa pressure for 10 min: diamond  $(\bullet)$ ;  $\alpha$ -iron  $(\circ)$ ; Fe<sub>3</sub>C ( $\circ$ ); Fe<sub>7</sub>C<sub>3</sub> ( $\circ$ ).



*Figure 7* Changes in relative amounts of crystalline phases formed by the mixture of 80 wt% graphite-20 wt% manganese under 7 GPa pressure for 10 min: diamond  $(\bullet)$ ;  $\alpha$ -manganese ( $\circ$ ); M<sub>7</sub>C<sub>3</sub> ( $\triangle$ ).

then Fe<sub>3</sub>C changed into Fe<sub>7</sub>C<sub>3</sub> by further reaction. The diamond was found to form at  $1500^{\circ}$ C and  $Fe<sub>3</sub>C$  was reformed at higher temperature. The yield of diamond was about 50 wt  $%$  in the specimen treated at 1700 $^{\circ}$ C. The behaviour of Fe<sub>3</sub>C and Fe<sub>7</sub>C<sub>3</sub> depends on pressure, so that detailed study of the behaviour of iron carbides in relation to diamond formation is necessary. Under 7 GPa pressure, the iron carbide  $Fe<sub>7</sub>C<sub>3</sub>$  was found to melt incongruently and diamond might form from iron-carbon solution above the peritectic temperature of Fe<sub>7</sub>C<sub>3</sub>. The Curie temperature of Fe<sub>3</sub>C was 210 $^{\circ}$ C and that of Fe<sub>7</sub>C<sub>3</sub> was 250 $^{\circ}$ C in agreement with reported values [6].

## 3.4. Diamond formation in manganese-graphite system

In the manganese-graphite system, only one stoichiometric carbide  $Mn<sub>7</sub>C<sub>3</sub>$  was formed at 7 GPa, although  $Mn_{23}C_6$  is a stable phase under atmospheric pressure [7]. The relative amounts of the crystalline phases is shown in Fig. 7. Manganese reacted with graphite to form  $Mn<sub>7</sub>C<sub>3</sub>$  at around  $1000^{\circ}$  C. The diamond formation took place at 1500 $^{\circ}$  C. The yield of diamond was about 20 wt % in the specimen at 1700 $^{\circ}$ C. Mn<sub>7</sub>C<sub>3</sub> did not change to the other kind of carbide in the entire temperature range used in this work. Giardini and Tydings [1] proposed that MnC, which was richer in carbon than  $Mn<sub>7</sub>C<sub>3</sub>$ , was formed and decomposed to

 $Mn<sub>7</sub>C<sub>3</sub>$  and diamond. In this work any indication of the formation of MnC was not observed. Mn<sub>7</sub>C<sub>3</sub> probably melts congruently, so diamond should be precipitated from the manganese-graphite solution above the  $Mn<sub>7</sub>C<sub>3</sub>$ -diamond eutectic temperature. A carbide poorer in carbon than  $Mn<sub>7</sub>C<sub>3</sub>$  may exist, though it was not detected in this work with the carbon-rich composition.

### **4. Summary**

Diamond was synthesized from the mixture of graphite and metal powder of nickel, cobalt, iron or manganese under 7GPa pressure. The formation and behaviour of their carbides were found to be classified into two groups. Nickel and cobalt form only nonstoichiometric carbides  $M<sub>r</sub>C$  with graphite; on the other hand, iron and manganese form stoichiometric carbides. In the former, the metal-carbon system have only one eutectic point (liquid  $\rightarrow$  diamond + M<sub>x</sub>C). Among the latter, manganese has stable carbide  $Mn<sub>7</sub>C<sub>3</sub>$ . But iron, which forms easily metastable  $Fe<sub>3</sub>C$  under atmospheric pressure, has two carbides of  $Fe<sub>3</sub>C$  and Fe<sub>7</sub>C<sub>3</sub> at 7 GPa pressure. The stability range of  $Fe<sub>7</sub>C<sub>3</sub>$  is not so wide that the change of carbide phase from  $Fe<sub>7</sub>C<sub>3</sub>$  to  $Fe<sub>3</sub>C$  could be detected at the temperature where diamond is formed. These differences certainly reflect the difference in the interactions between metal and carbon, which are related to the number of 3d-electrons in metals.

#### **R eferences**

- 1. A. A. GIARDINI and J. E. TYDINGS, *Amer. Mineral.*  47 (1962) 1393.
- 2. L.H. COHEN, W. KLEMENT and G.C. KENNEDY, *Phys. Rev.* 145 (1966) 519.
- 3. H. L. D. PUGH, J. LEES and J.A. BLAND, *Nature*  191 (1961) 865.
- 4. H.M. STRONG and R.E. HANNEMAN, *J. Chem. Phys.* 46 (1967) 3668.
- 5. H. KATZMAN and W. F. LIBBY, *Science* 172 (1971) 1132.
- 6. H. C. EKSTROM and W. A. ADOCK, *J. Amer. Chem. Soc.* 72 (1950) 1042.
- 7. M. HANSEN and K. ANDERKO, "Constitution of Binary Alloys" (McGraw-Hill Book Company Inc., New York, 1958).

*Received 1 February and accepted 18 May 1983*